



國家同步輻射研究中心年度實驗報告

NSRRC Annual Experiment Report

Form: ER1

計劃編號 (Proposal no.)	2011-1-021		所有光束線編號 (All BL no.)	BL13A, BL17B1		
計畫題目 (Proposal title)	High quality ZnO epitaxial films					
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High quality ZnO epitaxial films were grown by pulsed-laser deposition on Si(111) substrates with a thin MBE grown Y_2O_3 buffer layer.[1] Radial scan along surface normal of the sample of a 0.2 μ m thick ZnO layer, not shown, reveals c-plane oriented ZnO layer was observed on the Si(111) substrate with the cubic Y_2O_3 beffer layer also (111) oriented. The azimuthal ϕ -scans across non-specular reflections of $ZnO\{10-11\}$, $Y_2O_3\{440\}$ and $Si\{220\}$ reflections, shown in Fig. 1, were performed and yielded the in-plan epitaxial relationship of $ZnO\{10-10\}||Y_2O_3\{22-4\}||Si\{4-2-2\}$. Cubic Y_2O_3 has a bixbyite structure, which can be described as a vacancy-ordered fluorite. Viewing along the [111] direction of Y_2O_3 , the O sub-lattice in Y_2O_3 consists of two-dimensional defective hexagonal lattices stacking with ABC sequence along the [111] direction, as shown in Fig. 2(a), in which the filled circles denote O atoms and open circles represent O vacancies. The hexagonal unit cell has a lattice constant equal to $a(Y_2O_3) \cdot \sqrt{2}/4 = 3.750 \text{ \AA}$ and its axes are aligned with the $Y_2O_3\langle 10-1 \rangle$ directions, identical to the axes in ZnO basal plane. This elucidates the ZnO lattice is aligned with the O sub-lattice in Y_2O_3 , as illustrated in Fig. 2(b). The lattice mismatch between ZnO and O sub-lattices in Y_2O_3 and in sapphire are -13.5% and 18.1%, respectively. For systems with such a large lattice mismatch, the well established lattice matching epitaxy (LME), where films grow by one-to-one matching of lattice constants or pseudomorphically across the film-substrate interface, is not the favorable mechanism. Instead, domain matching epitaxy (DME) [2], where integral multiples of lattice planes containing densely packed rows are matched across the interface, provides a nice description of the interfacial structure of these systems. The planar spacing ratio of $ZnO(11-20)$ to parallel $Y_2O_3(4-40)$, which coincides with the (11-20) planes of O sub-lattice in Y_2O_3 falls between 6/7 and 7/8; this implies a matching of 7(8) planes of ZnO with 6(7) planes of Y_2O_3 across the interface along this direction. The large lattice mismatch is thus accommodated by the misfit dislocations localized at the interface.

References

[1] W.-R. Liu, Y.-H. Li, W. F. Hsieh, C.-H. Hsu, W. C. Lee, Y. J. Lee, and M. Hong, and J. Kwo,

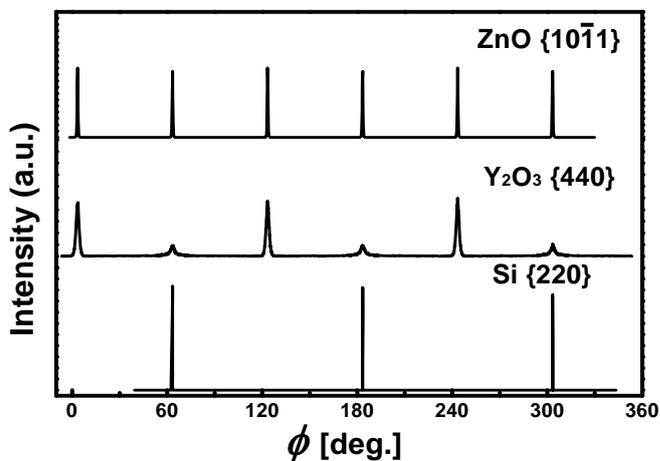


Fig. 1 ϕ -scan profiles across ZnO $\{10\bar{1}1\}$, $Y_2O_3\{440\}$, and Si $\{220\}$ off-normal reflections.

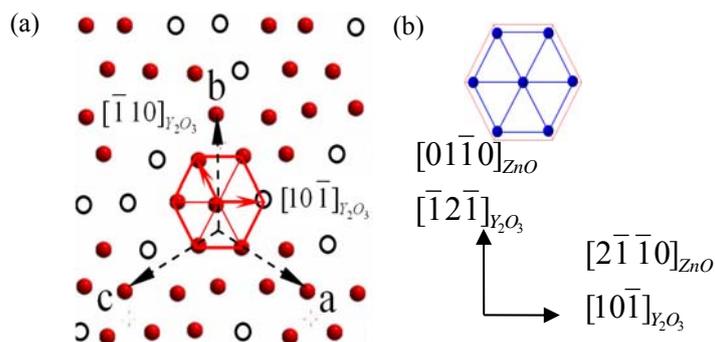


Fig. 2 (a) Schematic of atomic arrangement of O sub-lattice in Y_2O_3 (111) planes, where the filled circles are O atoms and the open circles denote O vacancies. The dashed arrows are (111) projection of the basis vectors of Y_2O_3 cubic lattice. (b) Illustration of the lattice alignment of ZnO basal plane (small hexagon) with O sub-lattice in Y_2O_3 (large hexagon).

